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**A DFT STUDY ON THE STRUCTURAL AND ELECTRONIC PROPERTIES OF  
SMALL TOXIC GASES ON B- AND Al-DOPED C<sub>20</sub> FULLERENE**

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The structural and electronic properties of semiconducting BC<sub>19</sub> and AlC<sub>19</sub> heterofullerenes as adsorbents for toxic small gas molecules (H<sub>2</sub>S and SO<sub>2</sub>) are determined by DFT. Structural parameters, energy gaps, natural population analysis, partial density of state, dipole moments, and vibrational frequencies were extracted. The adsorption process and sensitivity to the gases are increased by doping with B or Al. The results show that AlC<sub>19</sub> is the most sensitive structure. The good sensing of AlC<sub>19</sub> is related to high charge transfer upon gas adsorption. Adsorption of the H<sub>2</sub>S on the BC<sub>19</sub> has negligible effects on the electronic properties, to be categorized as "harmless adsorption". H<sub>2</sub>S is weakly adsorbed on BC<sub>19</sub> and AlC<sub>19</sub>. The H<sub>2</sub>S and SO<sub>2</sub> molecules act as electron donating and electron withdrawing molecules, respectively. Notably, the adsorption processes are highly exothermic. In general, BC<sub>19</sub> is more reactive than C<sub>20</sub> and AlC<sub>19</sub> is the most reactive cage. This provides a theoretical basis to fabricate B- and Al-doped C<sub>20</sub>-based gas sensors.

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**INTRODUCTION**

According to Euler's theorem, fullerenes are molecules that made up of exactly 12 pentagons and a differing number of hexagons [ 1 ]. In 1985, C<sub>60</sub> as one of the most famous fullerenes with 12 pentagons and 20 hexagons was discovered [ 2 ]. The fullerenes are promising candidates for basic elements in nanoscale systems. The C<sub>20</sub> molecule with only 12 pentagons is the smallest member of the fullerene family [ 3 ]. After detection of the C<sub>20</sub> in different phases [ 3–5 ], many theoretical and experimental studies on this smallest carbon cluster have been performed [ 6–17 ]. The C<sub>20</sub> fullerenes with different symmetries have very close energies and similar properties [ 9, 15, 17 ]. In addition, thermal stability and strong reactivity of C<sub>20</sub> have been confirmed [ 16, 17 ]. Despite the many studies on the C<sub>20</sub>, small effort has been made to assess the C<sub>20</sub> fullerene as a potential agent for adsorption of the small gases, even though many studies have been performed on potential applications of nanostructures for gas adsorption [ 16, 18–30 ]. One of the main challenges in the applied usage of the nanostructures as gas sensors is poor sensitivity. Nowadays, sensing properties of doped fullerenes, as a family of nanostructures, is one of the most interesting subjects [ 14, 31, 32 ]. Previous studies showed that doped fullerenes have strong affinity toward the small gases [ 31, 32 ]. The electronic and gas adsorption properties of graphene, CNT, BNNT, B<sub>12</sub>, and C<sub>60</sub> structures have been seriously affected by B and Al doping [ 18–23, 31, 33–35 ].

Various impurities in the petroleum industry and important atmospheric processes such as development of acid rain and ozone depletion are consist of sulfur-containing molecules [36, 37]. Several studies have focused on designing adsorbents for removing H<sub>2</sub>S and SO<sub>2</sub> from gases [24, 25, 28, 29, 38–41].

In this study, we report the first-principles calculations on the interactions between H<sub>2</sub>S and SO<sub>2</sub> with C<sub>20</sub> fullerene. The sensitivity of the heterofullerens toward the gases is analyzed.

### COMPUTATIONAL DETAILS

DFT calculations were carried out using the GAMESS package [42] with B3LYP method at 6-31G++(d,p) basis set. Geometries of all the systems without any constraints were fully optimized. The compatibility of B3LYP with the fullerene-based nanostructures has been confirmed [43, 44]. Energy, natural population analysis (NPA) [45], and partial density of states (PDOS) of the optimized structures are also investigated to extract the conclusions on the stability, charge transfer (CT), and electronic structures. The corrected adsorption energy ( $E_{\text{ads}}$ ) is given by:

$$E_{\text{ads}} = E_{\text{cage-gas}} - E_{\text{cage}} - E_{\text{gas}} + E_{\text{BSSE}}, \quad (1)$$

where  $E_{\text{cage-gas}}$ ,  $E_{\text{cage}}$ , and  $E_{\text{gas}}$  are the energy of the gas on the cage, the fully relaxed cage, and the isolated gas molecule, respectively. Due to the unbalancing between the basis set used to describe the interacting system and the isolated reactants,  $E_{\text{BSSE}}$  is applied to correct the deviations of the adsorption energy [46]. According to Pearson, operational and approximate definitions of the electronic chemical potential ( $\mu$ ) and the absolute hardness ( $\eta$ ) of a chemical system are given by [47]:

$$\mu \cong (-1/2)(\text{IP} + \text{EA}), \quad (2)$$

$$\eta = (1/2)(\text{IP} - \text{EA}), \quad (3)$$

where IP and EA are the ionization potential and electron affinity, respectively. According to Koopman's theorem, the frontier orbital energies are given by [48]

$$\text{IP} = -E_{\text{HOMO}}, \quad (4)$$

$$\text{EA} = -E_{\text{LUMO}}. \quad (5)$$

Also, global softness ( $S$ ) and electrophilicity index ( $\omega$ ) are computed from the orbital energies of the highest occupied molecular orbital ( $E_{\text{HOMO}}$ ) and the lowest unoccupied molecular orbital ( $E_{\text{LUMO}}$ ) at the same level of energy.

$$S = 1/2\eta, \quad (6)$$

$$\omega = \mu^2/2\eta. \quad (7)$$

The energy gap variation ( $\Delta E_g$ ) for any cage-gas complex is defined as

$$\Delta E_g = E_{g(\text{cage-gas})} - E_{g(\text{cage})}, \quad (8)$$

where  $E_g$  is differences between the highest occupied molecular orbital (HOMO) and the lowest unoccupied molecular orbital (LUMO). It should be noted that for systems with an odd number of total electrons, there are two distinct  $\alpha$  and  $\beta$   $E_g$ .

### RESULTS AND DISCUSSION

**The structural and electronic properties of C<sub>20</sub>, BC<sub>19</sub>, and AlC<sub>19</sub>.** First of all the C<sub>20</sub> fullerene without any constraints has been optimized and shown in Fig. 1, *a*. The structural and electronic properties of the cages are listed in Tables 1 and 2. The average bond length, bond angle, and diameter of the C<sub>20</sub> are about 1.44 Å, 108.72°, and 3.99 Å, respectively, which is in good agreement with the reported data [10, 12, 14, 16]. The cohesive energy ( $E_{\text{coh}}$ ) as a good criterion for thermodynamic stability of the structure is calculated from the formula

$$E_{\text{coh}} = E_{\text{cage}} - n_C E_C - n_{\text{Al}} E_{\text{Al}} - n_B E_B, \quad (9)$$

where  $E_{\text{cage}}$  is the total energy of C<sub>20</sub>, BC<sub>19</sub> or AlC<sub>19</sub>, and  $E_C$ ,  $E_B$ , and  $E_{\text{Al}}$  are the energy of an isolated C, B, and Al atom, respectively, and  $n_C$ ,  $n_B$ , and  $n_{\text{Al}}$  are the number of corresponding atoms. The total energy of the C<sub>20</sub> is calculated to be –761.602 Hartree, in good agreement with previous QMC and

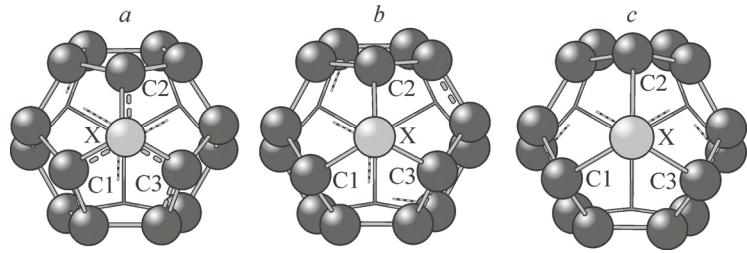


Fig. 1. The optimized structures of  $C_{20}$  (a),  $BC_{19}$  (b), and  $AlC_{19}$  (c)

DFT calculations [49, 50]. Larger absolute  $E_{coh}$  corresponds to more stability. Based on  $E_{coh}$ ,  $C_{20}$  fullerene ( $E_{coh} = -7.84$  eV/atom) is more reactive than both  $C_{60}$  ( $E_{coh} = -8.66$  eV/atom) and  $AlC_{59}$  ( $-8.51$  eV/atom) at the same level of theory. This is consistent with the previous studies [49—51]. Therefore,  $C_{20}$  is more useful than the  $C_{60}$  for further applications, and to calculate gas adsorption on the native and modified  $C_{20}$  fullerene can be of interest. The results show that  $H_2S$  and  $SO_2$  molecules are adsorbed with  $E_{ads}$  of  $-0.04$  and  $-0.2$  eV, respectively. Furthermore, the minimum distance ( $R_{min}$ ) between  $C_{20}$  and  $H_2S$  or  $SO_2$  molecule are  $3.9$  and  $2.52$  Å, respectively. The adsorptions processes are exothermic, but the values of  $E_{ads}$  are not suitable for practical usage. One of the proposed methods to overcome the insensitivity is doping of the  $C_{20}$  by B or Al atoms. The dopant acts as an adsorption site for the gas molecules and induces CT between the cage and the adsorbed molecules, making the doped cage capable of sensing  $H_2S$  and  $SO_2$ . Fig. 1, b, c shows the fully relaxed geometries of the heterofullerenes. Important changes in the bond lengths, bond angles, and diameter near the heteroatom have been observed. The average bond lengths of B—C and Al—C are  $1.56$  Å and  $1.93$  Å, respectively. Moreover, average bond angles of C—B—C and C—Al—C are  $106.97$  and  $93.26^\circ$ , indicating  $sp^3$  hybridization of the heteroatom. It is notable that, as revealed in Table 1, the cage diameter that passes through the heteroatom increases from  $3.99$  Å in the  $C_{20}$  to  $4.23$  Å and  $4.76$  Å in  $BC_{19}$  and  $AlC_{19}$ , respectively. Upon structural relaxation, the heteroatom is pushed outward from the surface and is ready to react with other molecules. So, protrusion of Al atom in  $AlC_{19}$  ( $0.77$  Å) is larger than that of B atom in  $BC_{19}$  ( $0.24$  Å), which can be described by atomic radius of heteroatom relative to that of the C atom [20, 52]. The protrusion of Al atom in CNT and  $C_{60}$  were calculated to be  $0.812$  and  $0.76$  Å, respectively [19]. The corresponding of the  $E_{coh}$  of  $BC_{19}$  and  $AlC_{19}$  are  $-7.75$  and  $-7.55$  eV/atom, respectively.

Table 1

Structural parameters of  $C_{20}$ ,  $BC_{19}$ , and  $AlC_{19}$  cages

System	Bond length, Å			Bond angle, deg.			Diameter, Å	Protrusion, Å
	C1—X	C2—X	C3—X	C1—X—C2	C1—X—C3	C2—X—C3		
$C_{20}$	1.44	1.44	1.44	108.73	108.73	108.71	3.99	—
$BC_{19}$	1.56	1.56	1.56	108.21	108.16	104.55	4.23	0.24
$AlC_{19}$	1.93	1.93	1.94	94.80	90.25	94.73	4.76	0.77

Table 2

Cohesive energy ( $E_{coh}$ ), band gap ( $E_g$ ), ionization potential (IP), electrophilicity index ( $\omega$ ), chemical potential ( $\mu$ ), hardness ( $\eta$ ), softness (S) and dipole moment ( $\mu_D$ ) of  $C_{20}$ ,  $BC_{19}$ , and  $AlC_{19}$  cages.  $\alpha$  and  $\beta$  denote spin up and down, respectively

System	$E_{coh}$ , eV/atom	$E_g$ , eV	IP, eV	$\omega$ , eV	$\mu$ , eV	$\eta$ , eV	S, eV	$\mu_D$ , D
$C_{20}$	-7.84	1.91	5.536	10.03	-4.580	0.956	0.523	0.00
$BC_{19}$	-7.74	$1.86^\alpha, 3.15^\beta$	5.478	9.615	-4.548	0.930	0.538	0.27
$AlC_{19}$	-7.55	$1.60^\alpha, 2.7^\beta$	5.248	7.908	-4.448	0.800	0.626	4.08

Table 3

*Natural population analysis (NPA) and natural atomic orbital occupancies of heteroatoms in free and in the cage*

System	NPA	2s	2p <sub>x</sub>	2p <sub>y</sub>	2p <sub>z</sub>	3s	3p <sub>x</sub>	3p <sub>y</sub>	3p <sub>z</sub>
B	0.00	2.00	1.00	0.00	0.00	0.00	0.00	0.00	0.00
BC <sub>19</sub>	+0.71	0.64	0.46	0.55	0.64	0.00	0.00	0.00	0.00
Al	0.00	2.00	2.00	2.00	2.00	2.00	1.00	0.00	0.00
AlC <sub>19</sub>	+1.63	2.00	2.00	2.00	2.00	0.68	0.14	0.28	0.26

tively. Thus, AlC<sub>19</sub> must be more reactive than BC<sub>19</sub> and C<sub>20</sub> structures. The same trend was also observed for AlC<sub>59</sub>, BC<sub>59</sub>, and C<sub>60</sub> fullerenes [ 53 ].

NPA and natural atomic orbital occupancies of the heteroatoms are summarized in Table 3. After doping the B atom loses 1.36 |e| and gains 0.63 |e| in 2s and 2p orbitals, respectively. Therefore, both donation and back-donation of electron density occurs. On the other hand, the Al atom loses 1.32 |e| and 0.32 |e| of its charge from 3s and 3p orbitals, respectively. This finding shows that the 3s orbital has a stronger interaction with the neighboring atoms than the 3p orbital. In both cases, the p<sub>x</sub> orbital loses electron density, whereas the p<sub>y</sub> and p<sub>z</sub> orbitals get more of it. As it can be seen from Table 3, there is an obvious electron deficient region on the B and Al atoms, consistent with a considerable positive electrostatic charge +0.71 |e| and +1.63 |e| for B and Al atoms, respectively. This suggests that the doping with Al and B atom forms electronic holes in the fullerene cage.

Chemical reaction for the reactant molecules can also be explained based on frontier molecular orbitals (FMO). PDOS of the C<sub>20</sub>, BC<sub>19</sub>, and AlC<sub>19</sub> are depicted in Fig. 2. Calculated E<sub>g</sub> of the cages are listed in Table 2. E<sub>g</sub> for C<sub>20</sub> is 1.91 eV. Many different results were reported for E<sub>g</sub> of C<sub>20</sub> [ 6, 7, 12, 16, 17 ]. These deviations can be attributed to the choice of the functional and variety of methods. PBE and B3LYP functional give different results, and the results from PBE are generally underestimated [ 17 ]. The geometrical changes would result in different electronic properties of the cages. Electronic structure analysis demonstrated that energy of HOMO level is more sensitive than LUMO level to the doping. In both BC<sub>19</sub> and AlC<sub>19</sub>, the HOMO levels shift towards more stable states, but this change is not visualized for the LUMO levels. As a result of removing peak in -5.6 eV, E<sub>g</sub> for the β level of BC<sub>19</sub> and AlC<sub>19</sub> are changed to 3.15 and 2.7 eV, respectively. Notably that the α gap of BC<sub>19</sub> and AlC<sub>19</sub> is smaller than β one. Therefore, the electronic properties of the systems are determined by the α level. The electrical conductivity of C<sub>20</sub> should be increased upon B- and Al-doping, which is consistent with the NPA results. Abundance of fullerene structures is related to both thermodynamic and kinetic stability. Since electron transition from HOMO to LUMO is energetically inappropriate, the structures with larger E<sub>g</sub> have higher kinetic stability and low reactivity [ 54 ]. Thus the kinetic stability of the cage is decreased upon substitution. Therefore, the doped cage with smaller heteroatom is more stable. The same trends for η, ω, E<sub>coh</sub>, and E<sub>g</sub> have also been observed. As listed in Table 2, BC<sub>19</sub> and AlC<sub>19</sub> have smaller IP, which confirms that the new cages can easily lose electrons. The tendency of a system to obtain additional charge and a measure of the electrophilic power of a molecule is described by the quantity of ω (Table 2). When two molecules react with each other, one of them acts as nucleophile while the other one acts as electrophile. There is a direct relationship between ω and electrophilic nature of a molecule. Based on ω, AlC<sub>19</sub> loses more electrons than BC<sub>19</sub> and C<sub>20</sub>. Thus, AlC<sub>19</sub> and BC<sub>19</sub> act as nucleophilic species. Increasing in η, which is like a stamina toward the deformation in the presence of an electric field, leads to an increase in the stability, and reduces the reactivity of the chemical species [ 55 ]. The S parameter is another index which has an inverse relationship with η. Therefore, AlC<sub>19</sub> has a maximum instability and reactivity respect to the other cages. From the above discussion, the reactivity trend is AlC<sub>19</sub> > BC<sub>19</sub> > C<sub>20</sub>.

The B and Al elements are the active site and the sensing element of the modified cages. In continuation of the work, the structures and energies of minimum adsorption configurations of H<sub>2</sub>S and SO<sub>2</sub> molecules with various possible initial adsorption geometries were investigated. For each adsor-

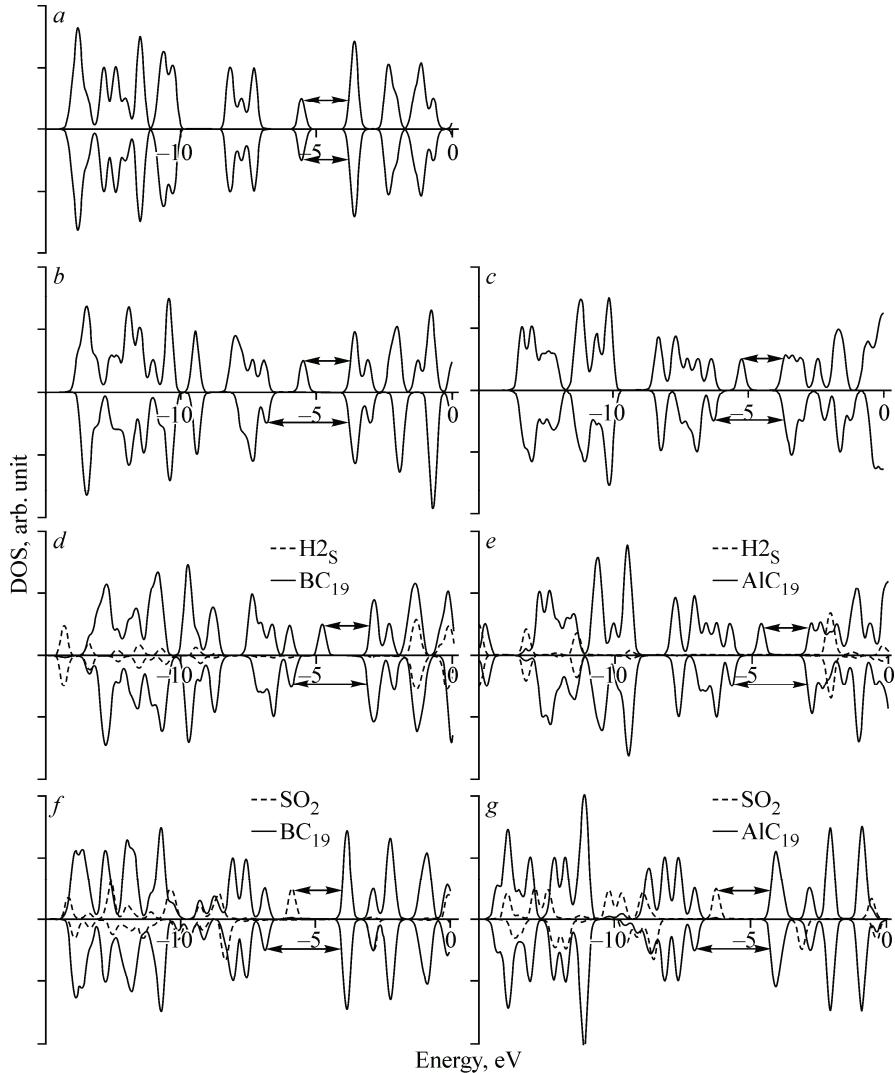


Fig. 2. Partial density of states for  $C_{20}$  (a),  $BC_{19}$  (b),  $AlC_{19}$  (c),  $H_2S@BC_{19}$  (d),  $H_2S@AlC_{19}$  (e),  $SO_2@BC_{19}$  (f), and  $SO_2@AlC_{19}$  (g).

The upper and lower panels are PDOS for spin up and spin down, respectively. The band gaps are shown by the arrows

bate, two adsorption sites, namely, on the top of a heteroatom and the center of pentagon with an impurity atom, with different configurations are evaluated.

**$H_2S$  on fullerenes.** The most stable configurations of  $H_2S$  adsorbed on the heterofullerenes are shown in Fig. 3, a, c. Tables 4–6 summarize the calculated structural and electronic parameters for the most optimized configuration of cage- $H_2S$  complex. In these configurations,  $R_{min}$  between B and Al atoms and the S atom are 2.05 and 2.48 Å, respectively. CT pathway can be explained by interactions between the HOMO of one species and the LUMO of the other one. As can be seen from Table 5,  $|LUMO(H_2S) - HOMO(\text{cage})|$  is larger than  $|LUMO(\text{cage}) - HOMO(H_2S)|$ . Thus, the electrons transfer from  $H_2S$  to cages is due to a small energy barrier. This interpretation is consistent with the findings from NPA. According to the NPA (Table 4), the  $H_2S$  molecule loses 0.53 |e| and 0.32 |e| on  $BC_{19}$  and  $AlC_{19}$ , respectively. The positive sign indicates the electron transfer from  $H_2S$  to the cages. The same behavior is observed between  $H_2S$  and the other adsorbents [24, 28–30]. The S–H bond lengths increase from 1.348 Å in isolated  $H_2S$  to 1.352 Å and 1.353 Å for S–H1 and S–H2 bonds on both heterofullerenes (Table 6). As a result of CT from bonding orbital, the elongation of the bond lengths is justified. In addition, the H–S–H bond angle is increased from 92.472° to 93.386 and

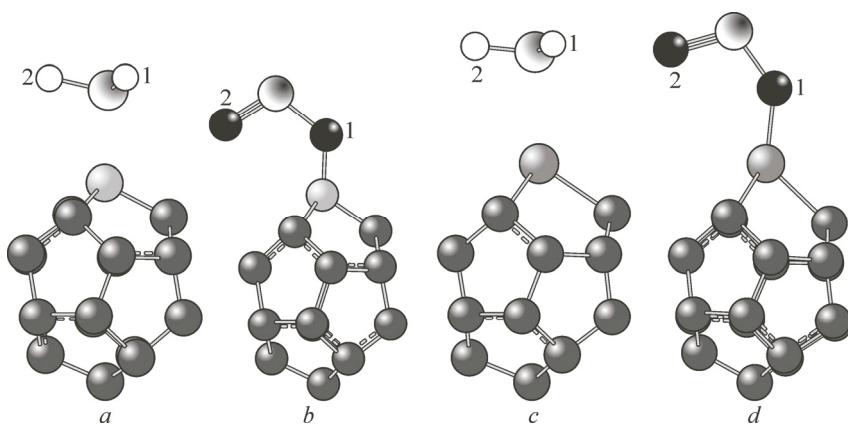
Fig. 3. The optimized structures of  $\text{H}_2\text{S}@\text{BC}_{19}$  (a),  $\text{SO}_2@\text{BC}_{19}$  (b),  $\text{H}_2\text{S}@\text{AlC}_{19}$  (c),  $\text{SO}_2@\text{AlC}_{19}$  (d)

Table 4

*Structural and electronic properties of the most stable form of gas-cage system. Including minimum distance of cage-gas ( $R_{\min}$ ), amount of difference of protrusion before and after gas adsorption ( $\Delta P$ ) ( $\Delta P = P_{\text{after}} - P_{\text{before}}$ ), band gap ( $E_g$ ), the amount of difference of  $E_g$  before and after adsorption ( $\Delta E_g$ ), amount of CT from gas to cage ( $Q_T$ ), amount of charge on B or Al atom in the complexes from NPA ( $Q_X$ ) ( $X = \text{B}$  or  $\text{Al}$ ), dipole moment ( $\mu_D$ ), and corrected adsorption energy ( $E_{\text{ads}}$ )*

System	$R_{\min}$ , Å	$\Delta P$ , Å	$E$ , eV	$\Delta E_g$ , eV	$Q_T$ ,  e	$Q_X$ ,  e	$\mu_D$ , D	$E_{\text{ads}}$ , eV
$\text{BC}_{19}-\text{H}_2\text{S}$	2.05	0.2	1.85 <sup>a</sup> , 2.90 <sup>b</sup>	-0.01 <sup>a</sup> , -0.25 <sup>b</sup>	+0.53	0.33	6.93	-0.43
$\text{BC}_{19}-\text{SO}_2$	1.44	0.39	2.00 <sup>a</sup> , 3.02 <sup>b</sup>	+0.14 <sup>a</sup> , -0.13 <sup>b</sup>	-0.49	0.63	2.75	-1.7
$\text{AlC}_{19}-\text{H}_2\text{S}$	2.48	0.07	1.84 <sup>a</sup> , 2.87 <sup>b</sup>	+0.24 <sup>a</sup> , +0.17 <sup>b</sup>	+0.32	1.41	10.18	-0.91
$\text{AlC}_{19}-\text{SO}_2$	1.75	0.3	2.21 <sup>a</sup> , 3.03 <sup>b</sup>	+0.61 <sup>a</sup> , +0.33 <sup>b</sup>	-0.69	1.75	2.62	-3.03

Table 5

*Electrical structure parameters of the adsorption structures*

Molecule	$E(\text{HOMO})$	$E(\text{LUMO})$	$ E_{\text{LUMO}}(\text{cage}) - E_{\text{HOMO}}(\text{gas}) $	$ E_{\text{LUMO}}(\text{gas}) - E_{\text{HOMO}}(\text{cage}) $
$\text{BC}_{19}$	-5.48	-3.62		
$\text{H}_2\text{S}$	-7.30	-0.26	3.68	5.22
$\text{SO}_2$	-9.46	-4.11	5.84	1.37
$\text{AlC}_{19}$	-5.25	-3.65		
$\text{H}_2\text{S}$	-7.30	-0.26	3.65	4.99
$\text{SO}_2$	-9.46	-4.11	5.75	1.14

Table 6

*Structural parameters of gas-cage complexes. X is denoted B or Al atoms*

System	Bond length, Å							Bond angle, deg.	
	C1—X	C2—X	C3—X	H1—S	H2—S	O1—S	O2—S	H1—S—H2	O1—S—O2
$\text{BC}_{19}-\text{H}_2\text{S}$	1.615	1.622	1.617	1.352	1.353			93.386	
$\text{AlC}_{19}-\text{H}_2\text{S}$	1.950	1.948	1.945	1.353	1.353			93.551	
$\text{BC}_{19}-\text{SO}_2$	1.668	1.670	1.654			1.501	1.648		111.073
$\text{AlC}_{19}-\text{SO}_2$	2.009	2.011	2.010			1.500	1.629		109.653

$93.551^\circ$  on  $\text{BC}_{19}$  and  $\text{AlC}_{19}$ , respectively. This increase can be attributed to the hindrance effect. The average C—X bond lengths in the heteropentagon ring increases from  $1.56\text{ \AA}$  to  $1.62\text{ \AA}$  and from  $1.93\text{ \AA}$  to  $1.95\text{ \AA}$  for  $\text{BC}_{19}$  and  $\text{AlC}_{19}$ , respectively. Large changes in  $\text{BC}_{19}$  are consistent with more CT toward  $\text{BC}_{19}$ . The charge on B and Al atoms are 0.33 and  $1.41|e|$ , respectively. The adsorptios of  $\text{H}_2\text{S}$  on the heterofullerenes is more exothermic than that on the fullerene. From the above analysis, the  $\text{H}_2\text{S}$  interaction with the  $\text{BC}_{19}$  is stronger than with  $\text{C}_{20}$ , and the strongest interaction is with  $\text{AlC}_{19}$ . Furthermore, the  $\text{H}_2\text{S}$  interaction with  $\text{AlC}_{19}$  is stronger than with CNT [30], AlN nanotube [24], Fe(110) [56], Pd(111) [39],  $\text{BC}_3$  sheet [57], but weaker than with Au-doped CNT [28] and Si-doped CNT [30]. Therefore,  $\text{H}_2\text{S}$  can be effectively adsorbed on the surface of both  $\text{BC}_{19}$  and  $\text{AlC}_{19}$ . Hence, both cages can be used for adsorption.

To elucidate further the adsorption processes of  $\text{H}_2\text{S}$ , the PDOSs plots of the cages and  $\text{H}_2\text{S}$  are shown in Fig. 2, *d*, *e*. Although a medium hybridization between the gas and cages in the region of  $-14$  to  $-9\text{ eV}$  and  $-3$  to  $0\text{ eV}$  occurs; the  $\text{H}_2\text{S}$  adsorption has no sensible effects on the electronic properties of  $\text{BC}_{19}$ , whereas the gap of the  $\alpha$  level on  $\text{AlC}_{19}$  increases by about  $0.24\text{ eV}$ . Therefore, the  $\text{AlC}_{19}$  conductivity decreases. As a result,  $\text{BC}_{19}$  cannot be a good sensor for  $\text{H}_2\text{S}$  gas, whereas  $\text{AlC}_{19}$  can.

**$\text{SO}_2$  on fullerenes.** After careful structural optimizations without any constraints, it is found that one of the oxygen atoms (O1) in  $\text{SO}_2$  is bonded to the heteroatom (Fig. 3, *b* and *d*). The structural and electronic properties of the most stable adduct in the cage- $\text{SO}_2$  system are summarized in Tables 4—6. In this configuration,  $R_{\min}$  between B and Al atoms to O1 of  $\text{SO}_2$  are  $1.44\text{ \AA}$  and  $1.75\text{ \AA}$ , respectively. The  $|\text{LUMO}(\text{SO}_2) - \text{HOMO}(\text{cage})|$  is smaller than  $|\text{LUMO}(\text{cage}) - \text{HOMO}(\text{SO}_2)|$  (Table 5). The electron needs to cross an energy barrier of more than  $5.7\text{ eV}$  to be transferred from  $\text{SO}_2$  to the cages. The NPA results are confirmed by this interpretation. The NPA shows that the  $\text{SO}_2$  molecule withdraw about  $0.49$  and  $0.69|e|$  charge from  $\text{BC}_{19}$  and  $\text{AlC}_{19}$ , respectively, owing to higher electronegativity of the O atoms. The same behavior is also observed with other adsorbents [25—28, 40, 58]. Because of the strong adsorption of  $\text{SO}_2$ , CT is remarkable. Influence of Al doping can be confirmed by large CT from  $\text{AlC}_{19}$  to  $\text{SO}_2$  gas (Table 4). Meanwhile, the electrons fill the antibonding orbital of  $\text{SO}_2$ . Thus, bond lengths increase from  $1.46\text{ \AA}$  in isolated  $\text{SO}_2$  to  $1.50\text{ \AA}$  and  $1.65\text{ \AA}$  for S—O1 and S—O2 bonds in the  $\text{BC}_{19}$ — $\text{SO}_2$ , and to  $1.50\text{ \AA}$  and  $1.63\text{ \AA}$  in the  $\text{AlC}_{19}$ — $\text{SO}_2$  system. The O—S—O angle decreases from  $118.7^\circ$  to  $111.1^\circ$  and  $109.6^\circ$  for  $\text{BC}_{19}$  and  $\text{AlC}_{19}$ , respectively. The significant structural changes in the heterofullerenes are explained by the strong attraction of  $\text{SO}_2$ . As a result of the C—X bond length elongation (Table 6), the bond energy and bond activity are decreased and increased, respectively. The B—C1, B—C2, B—C3 Al—C1, Al—C2, and Al—C3 bonds are elongated by 6.9, 7, 6.8, 4, 4.2, and 4.1 %, respectively. The  $\text{SO}_2$  molecule is adsorbed with the binding energy of  $-1.7$  and  $-3.03\text{ eV}$  on  $\text{BC}_{19}$  and  $\text{AlC}_{19}$ , respectively. Sensing of the gases requires relatively strong interactions between gas molecules and a sensor. The strong interaction of  $\text{SO}_2$  on  $\text{AlC}_{19}$  is not favorable for a gas sensor, but  $\text{BC}_{19}$  can be a good sensor for  $\text{SO}_2$ .  $E_{\text{ads}}$  on  $\text{AlC}_{19}$  is larger than that on B- and Al-doped graphene [22], Au-doped CNT [28], Pd-doped CNT [27], Ni-doped CNT [59], GeC nanotube [40], AlN nanotube and nanosheet [25, 58], and  $\text{Zn}_{12}\text{O}_{12}$  nanocage [26].

The PDOS plots are shown in Fig. 2. Upon adsorption of the  $\text{SO}_2$  molecule, great changes in PDOS happen. The HOMO and LUMO levels of the cages shift to lower energy states, while opposite shifting is evidenced for  $\text{H}_2\text{S}$  adsorption. After  $\text{SO}_2$  adsorption, the HOMO and LUMO of the  $\alpha$  level are exclusively constituted by the  $\text{SO}_2$  (100 %) and cage MO (100 %), respectively. The states of  $\text{SO}_2$  and cages overlap in the energy region of  $-14$  to  $-7\text{ eV}$ . An impurity peak can be seen in the region of  $-5.9$  and  $-6.34\text{ eV}$  for  $\text{BC}_{19}$ — $\text{SO}_2$  and  $\text{AlC}_{19}$ — $\text{SO}_2$  adducts. The  $E_g$  values of  $\text{BC}_{19}$ — $\text{SO}_2$  and  $\text{AlC}_{19}$ — $\text{SO}_2$  are larger by  $0.14$  and  $0.61\text{ eV}$  than those of free  $\text{BC}_{19}$  and  $\text{AlC}_{19}$ . This suggests that the cage- $\text{SO}_2$  adduct remains semiconductor in nature. Electron hopping over a large HOMO and LUMO gap is difficult and thus could diminish the conductance of the sensors. In conclusion,  $\text{BC}_{19}$  and  $\text{AlC}_{19}$  have a good sensitivity to  $\text{SO}_2$  and can be used to detect  $\text{SO}_2$  gas based on the changes in its conductivity. Hence, we propose that both  $\text{BC}_{19}$  and  $\text{AlC}_{19}$  may be suitable for detecting the presence of  $\text{SO}_2$  molecules.

**Electric dipole moment and vibrational frequencies.** The charge distribution in the structures is shown by electric  $\mu_D$  in Tables 2 and 4. Upon dopage,  $\mu_D$  of the heterofullerenes are increased. The structure with a large protrusion showed large  $\mu_D$ . The  $\mu_D$  values for BC<sub>19</sub> and AlC<sub>19</sub> are 0.27 and 4.08 D, respectively. For the H<sub>2</sub>S adducts with BC<sub>19</sub> and AlC<sub>19</sub> these values are 6.93 and 10.18 D, respectively. The sizes of the  $\mu_D$  vector of the cages are changed and increased by adsorption of the H<sub>2</sub>S molecule. Interestingly, the SO<sub>2</sub> gas molecule has different behavior on total  $\mu_D$  of the heterofullerenes. The size and direction of the electric  $\mu_D$  vector is changed. Upon SO<sub>2</sub> adsorption,  $\mu_D$  of the BC<sub>19</sub> and AlC<sub>19</sub> adducts as a criterion of charge distribution is increased and decreased, respectively. This contradiction is related to the polarization turning. It seems that there is a direct relationship between  $\mu_D$  of free molecules and  $E_{ads}$ . SO<sub>2</sub> and AlC<sub>19</sub> molecules with larger  $\mu_D$  values (2.01 and 4.08 D) have the largest amounts of  $E_{ads}$ , whereas H<sub>2</sub>S and BC<sub>19</sub> molecules with lower  $\mu_D$  (1.30 and 0.27 D, respectively) have smaller amounts of  $E_{ads}$ . The complexes with large  $\mu_D$  may have potential applications in dielectrics and solid state materials.

Local minima on the potential energy surface of the cages with and without the gases are confirmed by the vibrational frequency analysis. The calculated vibrational frequencies for all the cages show no imaginary vibrational frequency, indicating that the structures correspond to the true minima. The calculated infrared (IR) spectra are plotted in Fig. 4. The C<sub>20</sub> molecule has 54 frequency modes totally. From Fig. 4 we can see that C<sub>20</sub> has two distinct peaks at 730 and 1350 cm<sup>-1</sup>. As for the doped cages, their IR spectra are more complicated. The original two peaks are split and additional absorption is present. These absorptions all exhibit a red or blue shift as compared with those of the pristine cage. Furthermore, it can be seen that there are also several new peaks in the range of 250—500 cm<sup>-1</sup> for AlC<sub>19</sub>, while there are no such peaks for C<sub>20</sub> and BC<sub>19</sub>. As shown in Fig. 4, *d*, *e*, a sharp peak is observed at 686 and 404 cm<sup>-1</sup> for BC<sub>19</sub>—H<sub>2</sub>S and AlC<sub>19</sub>—H<sub>2</sub>S adducts, respectively. A small peak around 2650 cm<sup>-1</sup> for both heterofullerene adducts with H<sub>2</sub>S appears. Two strong peaks at 1040 and 1130 cm<sup>-1</sup> for BC<sub>19</sub>—SO<sub>2</sub>, and a distinct peak around 930 cm<sup>-1</sup> for AlC<sub>19</sub>—SO<sub>2</sub> can be used as finger

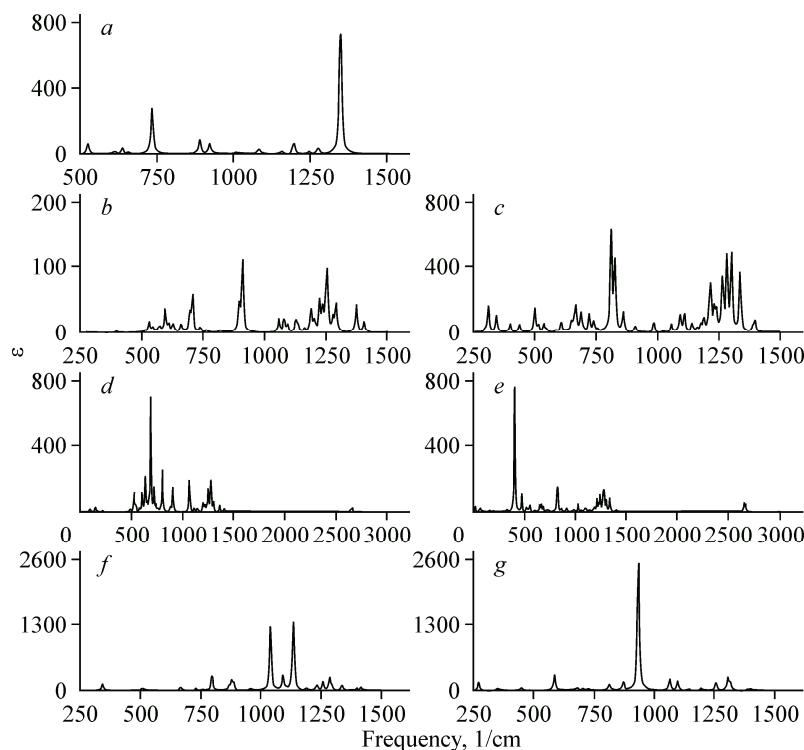


Fig. 4. The calculated IR spectra of the C<sub>20</sub> (*a*), BC<sub>19</sub> (*b*), AlC<sub>19</sub> (*c*), H<sub>2</sub>S@BC<sub>19</sub> (*d*), H<sub>2</sub>S@AlC<sub>19</sub> (*e*), SO<sub>2</sub>@BC<sub>19</sub> (*f*), and SO<sub>2</sub>@AlC<sub>19</sub> (*g*)

prints (Fig. 4, *f* and *g*). It can be seen that the shapes of IR absorption spectra are different for the different doped cages. These characteristic features in the IR spectra could be helpful to identify these heterofullerenes with and without gas molecules from the experimental spectra.

## CONCLUSIONS

DFT calculations were performed to study H<sub>2</sub>S and SO<sub>2</sub> gas adsorption on the BC<sub>19</sub> and AlC<sub>19</sub> heterofullerenes. After doping C<sub>20</sub> with B or Al, the local structure around the impurity region is affected. Furthermore, the C<sub>20</sub> conductivity was increased by doping. According to the calculated cohesive energies, the heterofullerene with a smaller heteroatom is more stable. The results show that BC<sub>19</sub> is not a good sensor for detecting H<sub>2</sub>S, while the electronic structure of AlC<sub>19</sub> is much affected by H<sub>2</sub>S adsorption. Both heterofullerenes are suitable sensors for SO<sub>2</sub>. Moreover, SO<sub>2</sub> has a profound impact on the electronic structure of the heterofullerenes and decreases their electrical conductivity. The C<sub>20</sub> reactivity toward H<sub>2</sub>S and SO<sub>2</sub> would be improved by doping with B and Al atoms. The IR absorptions exhibit red or blue shifts after the introduction of heteroatoms as compared with IR of pristine C<sub>20</sub>. We believe that the present work may help to develop new fullerene-based systems for gas sorption and sensing.

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